

## Complex Cubic $A_6B$ Compounds. I. The Crystal Structure of $Na_6Tl^*$

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$Na_6Tl$  represents one of the two structure types so far found for complex  $A_6B$  compounds of cubic symmetry containing about 400 atoms each per smallest unit cube, arranged according to space group  $F\bar{4}3m$ . A complete structure determination has been carried out for this compound, with the use of packing maps and subsequent least-squares refinements employing intensity data of 1602 reflections measured with an automated X-ray diffractometer. The length of the cube edge is  $a_0 = 24.154 \pm 0.001$  Å (Cu  $K\alpha_1 = 1.54051$  Å) giving a calculated density of  $\rho_c = 2.302$  g.cm<sup>-3</sup>. The final  $R$  value is 0.137. This unit cube contains one formula unit of  $Na_{3.52}Tl_{5.6}$  (or eight units of  $Na_{4.4}Tl_7$ ) and the atoms are distributed among 15 different point sets. Because of the likelihood of variable composition, especially in possible isostructural compounds, it seems practical to retain the formula  $Na_6Tl$  as a name for this structure type. The basic building block of the structure consists of a complex of 14 icosahedra and 42 pentagonal prisms, which is nearly identical to one of the four complexes occurring in  $Cu_4Cd_3$ . Four such complexes share a Laves-Friauf polyhedron, the center of which was not fully occupied in the crystal used. Each thallium atom is surrounded by an icosahedron of sodium atoms in such a way that there are no Tl-Tl contacts as was predicted on the basis of magnetic measurements.

### Introduction

The crystal structure study of  $Na_6Tl$  reported here was proposed to us by Professor J. F. Smith (Department of Metallurgy, Iowa State University) for the purpose of supplementing the results, obtained at his laboratory from measurements of magnetic susceptibilities. These were found to be temperature-dependent paramagnetic and of the Curie-Weiss type, giving a Curie constant of  $(4.92 \pm 0.25) \times 10^{-2}$  e.m.u., (°K/g.f.w.), which corresponds to  $0.63 \pm 0.02$  Bohr magnetons per formula unit of  $Na_6Tl$  (*i.e.* per Tl atom). On the basis of this result and other calculations, Greiner, Hansen, & Smith (1969) predicted that the thallium atoms in  $Na_6Tl$  are isolated from one another, implying a Tl-Tl internuclear separation of about 5 Å or more.

A complete crystal-structure determination ( $R = 16\%$ ) confirming the above prediction was carried out here, during the summer of 1967, with the use of X-ray data obtained from an irregular crystal at the Ames Laboratory. The results were reported to that laboratory, and an outline of some of the structural features together with the magnetic data was given by Greiner *et al.* (1969). The structure is cubic, space group  $F\bar{4}3m$ ,  $a_0 = 24.154$  Å.

The refinement of the structural parameters, and hence the present final report, was delayed to await the

collection of improved X-ray data from a spherically ground crystal and the development of facilities for sphere grinding of the extremely reactive compound.

Earlier investigations establishing the existence of  $Na_6Tl$ , but providing no crystallographic data, were those of Kurnakow & Pushin (1902) and Grube & Schmidt (1936). The two papers present almost identical results, stating that the compound forms peritectically at 85.9 at.% sodium ( $Na_{6.1}Tl$ ) and 78°C.

### Experimental

#### Sample preparation

A 10-gram sample of  $Na_6Tl$  (12.1 at.% Tl) was prepared by melting together reagent grade sodium (Fisher Scientific Company) and thallium of 99.99% purity (United Mineral and Chemical Corporation) in a stainless steel crucible, which was loaded and hermetically sealed by arc welding under dry argon gas. The sample was heated for one day at 350°C and frequently agitated, then annealed inside an oil bath for two days at  $70.0 \pm 0.5$ °C, for two days at  $66.0 \pm 0.5$ °C, and finally cooled to room temperature at a rate of about 3° per hour.

The crucible was opened inside a locally designed and constructed glove box, equipped with a sphere grinder and a continuously operating, very effective NaK ('nack') gettering device, thus allowing ample time to manipulate the extremely reactive single crystals without causing detectable corrosion. Details of the apparatus will be described elsewhere.

Because  $Na_6Tl$  is soft, spheres were obtained within a few minutes of grinding. These were encapsulated in glass capillaries according to the description given by

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Hansen (1968), thus avoiding the use of oil for corrosion protection.

Unit cell and space group

Laue, rotation, and Weissenberg photographs of layer lines 0 to 4 about [001] obtained with the use of an encapsulated sphere showed that the structure was face-centered cubic having Laue symmetry *m3m*. Only

reflections of the type *hkl*:  $h+k, k+l, (l+h)=2n$  were present. The probable space group is, accordingly, *F432*, *Fm3m* or *F43m*.

The length of the cube edge had been determined earlier at the Ames Laboratory with the use of an irregular crystal fragment, a General Electric diffractometer equipped with a 0.02° detector slit and a copper-target X-ray tube. Half-height,  $2\theta$  values were

Table 1. Observed and calculated structure factors for Na<sub>6</sub>Tl

Each group of four columns contains, from left to right, indices *l*, observed structure factors, calculated structure factors, and standard deviations. Two dashes indicate that the measured value was given zero weight. Values of *F*(cal) are based on the final isotropic refinement (Table 2).

l	Observed	Calculated	Standard Deviation
0 0 0	16 400 457 22	16 400 457 22	0
0 0 1	18 0 0 0	18 0 0 0	0
0 0 2	2 745 745 7	2 745 745 7	0
0 0 3	4 0 0 0	4 0 0 0	0
0 0 4	6 1306 1306 6	6 1306 1306 6	0
0 0 5	8 141 150 27	8 141 150 27	0
0 0 6	10 922 922 10	10 922 922 10	0
0 0 7	12 2149 2149 12	12 2149 2149 12	0
0 0 8	14 250 250 14	14 250 250 14	0
0 0 9	16 148 126 12	16 148 126 12	0
0 0 10	18 1210 1210 18	18 1210 1210 18	0
0 0 11	20 3 3 20	20 3 3 20	0
0 0 12	22 240 240 22	22 240 240 22	0
0 0 13	24 420 420 24	24 420 420 24	0
0 0 14	26 340 340 26	26 340 340 26	0
0 0 15	28 92 92 28	28 92 92 28	0
0 0 16	30 1410 1410 30	30 1410 1410 30	0
0 0 17	32 15 15 32	32 15 15 32	0
0 0 18	34 50 50 34	34 50 50 34	0
0 0 19	36 100 100 36	36 100 100 36	0
0 0 20	38 200 200 38	38 200 200 38	0
0 0 21	40 300 300 40	40 300 300 40	0
0 0 22	42 400 400 42	42 400 400 42	0
0 0 23	44 500 500 44	44 500 500 44	0
0 0 24	46 600 600 46	46 600 600 46	0
0 0 25	48 700 700 48	48 700 700 48	0
0 0 26	50 800 800 50	50 800 800 50	0
0 0 27	52 900 900 52	52 900 900 52	0
0 0 28	54 1000 1000 54	54 1000 1000 54	0
0 0 29	56 1100 1100 56	56 1100 1100 56	0
0 0 30	58 1200 1200 58	58 1200 1200 58	0
0 0 31	60 1300 1300 60	60 1300 1300 60	0
0 0 32	62 1400 1400 62	62 1400 1400 62	0
0 0 33	64 1500 1500 64	64 1500 1500 64	0
0 0 34	66 1600 1600 66	66 1600 1600 66	0
0 0 35	68 1700 1700 68	68 1700 1700 68	0
0 0 36	70 1800 1800 70	70 1800 1800 70	0
0 0 37	72 1900 1900 72	72 1900 1900 72	0
0 0 38	74 2000 2000 74	74 2000 2000 74	0
0 0 39	76 2100 2100 76	76 2100 2100 76	0
0 0 40	78 2200 2200 78	78 2200 2200 78	0
0 0 41	80 2300 2300 80	80 2300 2300 80	0
0 0 42	82 2400 2400 82	82 2400 2400 82	0
0 0 43	84 2500 2500 84	84 2500 2500 84	0
0 0 44	86 2600 2600 86	86 2600 2600 86	0
0 0 45	88 2700 2700 88	88 2700 2700 88	0
0 0 46	90 2800 2800 90	90 2800 2800 90	0
0 0 47	92 2900 2900 92	92 2900 2900 92	0
0 0 48	94 3000 3000 94	94 3000 3000 94	0
0 0 49	96 3100 3100 96	96 3100 3100 96	0
0 0 50	98 3200 3200 98	98 3200 3200 98	0
0 0 51	100 3300 3300 100	100 3300 3300 100	0
0 0 52	102 3400 3400 102	102 3400 3400 102	0
0 0 53	104 3500 3500 104	104 3500 3500 104	0
0 0 54	106 3600 3600 106	106 3600 3600 106	0
0 0 55	108 3700 3700 108	108 3700 3700 108	0
0 0 56	110 3800 3800 110	110 3800 3800 110	0
0 0 57	112 3900 3900 112	112 3900 3900 112	0
0 0 58	114 4000 4000 114	114 4000 4000 114	0
0 0 59	116 4100 4100 116	116 4100 4100 116	0
0 0 60	118 4200 4200 118	118 4200 4200 118	0
0 0 61	120 4300 4300 120	120 4300 4300 120	0
0 0 62	122 4400 4400 122	122 4400 4400 122	0
0 0 63	124 4500 4500 124	124 4500 4500 124	0
0 0 64	126 4600 4600 126	126 4600 4600 126	0
0 0 65	128 4700 4700 128	128 4700 4700 128	0
0 0 66	130 4800 4800 130	130 4800 4800 130	0
0 0 67	132 4900 4900 132	132 4900 4900 132	0
0 0 68	134 5000 5000 134	134 5000 5000 134	0
0 0 69	136 5100 5100 136	136 5100 5100 136	0
0 0 70	138 5200 5200 138	138 5200 5200 138	0
0 0 71	140 5300 5300 140	140 5300 5300 140	0
0 0 72	142 5400 5400 142	142 5400 5400 142	0
0 0 73	144 5500 5500 144	144 5500 5500 144	0
0 0 74	146 5600 5600 146	146 5600 5600 146	0
0 0 75	148 5700 5700 148	148 5700 5700 148	0
0 0 76	150 5800 5800 150	150 5800 5800 150	0
0 0 77	152 5900 5900 152	152 5900 5900 152	0
0 0 78	154 6000 6000 154	154 6000 6000 154	0
0 0 79	156 6100 6100 156	156 6100 6100 156	0
0 0 80	158 6200 6200 158	158 6200 6200 158	0
0 0 81	160 6300 6300 160	160 6300 6300 160	0
0 0 82	162 6400 6400 162	162 6400 6400 162	0
0 0 83	164 6500 6500 164	164 6500 6500 164	0
0 0 84	166 6600 6600 166	166 6600 6600 166	0
0 0 85	168 6700 6700 168	168 6700 6700 168	0
0 0 86	170 6800 6800 170	170 6800 6800 170	0
0 0 87	172 6900 6900 172	172 6900 6900 172	0
0 0 88	174 7000 7000 174	174 7000 7000 174	0
0 0 89	176 7100 7100 176	176 7100 7100 176	0
0 0 90	178 7200 7200 178	178 7200 7200 178	0
0 0 91	180 7300 7300 180	180 7300 7300 180	0
0 0 92	182 7400 7400 182	182 7400 7400 182	0
0 0 93	184 7500 7500 184	184 7500 7500 184	0
0 0 94	186 7600 7600 186	186 7600 7600 186	0
0 0 95	188 7700 7700 188	188 7700 7700 188	0
0 0 96	190 7800 7800 190	190 7800 7800 190	0
0 0 97	192 7900 7900 192	192 7900 7900 192	0
0 0 98	194 8000 8000 194	194 8000 8000 194	0
0 0 99	196 8100 8100 196	196 8100 8100 196	0
0 0 100	198 8200 8200 198	198 8200 8200 198	0
0 0 101	200 8300 8300 200	200 8300 8300 200	0
0 0 102	202 8400 8400 202	202 8400 8400 202	0
0 0 103	204 8500 8500 204	204 8500 8500 204	0
0 0 104	206 8600 8600 206	206 8600 8600 206	0
0 0 105	208 8700 8700 208	208 8700 8700 208	0
0 0 106	210 8800 8800 210	210 8800 8800 210	0
0 0 107	212 8900 8900 212	212 8900 8900 212	0
0 0 108	214 9000 9000 214	214 9000 9000 214	0
0 0 109	216 9100 9100 216	216 9100 9100 216	0
0 0 110	218 9200 9200 218	218 9200 9200 218	0
0 0 111	220 9300 9300 220	220 9300 9300 220	0
0 0 112	222 9400 9400 222	222 9400 9400 222	0
0 0 113	224 9500 9500 224	224 9500 9500 224	0
0 0 114	226 9600 9600 226	226 9600 9600 226	0
0 0 115	228 9700 9700 228	228 9700 9700 228	0
0 0 116	230 9800 9800 230	230 9800 9800 230	0
0 0 117	232 9900 9900 232	232 9900 9900 232	0
0 0 118	234 10000 10000 234	234 10000 10000 234	0
0 0 119	236 10100 10100 236	236 10100 10100 236	0
0 0 120	238 10200 10200 238	238 10200 10200 238	0
0 0 121	240 10300 10300 240	240 10300 10300 240	0
0 0 122	242 10400 10400 242	242 10400 10400 242	0
0 0 123	244 10500 10500 244	244 10500 10500 244	0
0 0 124	246 10600 10600 246	246 10600 10600 246	0
0 0 125	248 10700 10700 248	248 10700 10700 248	0
0 0 126	250 10800 10800 250	250 10800 10800 250	0
0 0 127	252 10900 10900 252	252 10900 10900 252	0
0 0 128	254 11000 11000 254	254 11000 11000 254	0
0 0 129	256 11100 11100 256	256 11100 11100 256	0
0 0 130	258 11200 11200 258	258 11200 11200 258	0
0 0 131	260 11300 11300 260	260 11300 11300 260	0
0 0 132	262 11400 11400 262	262 11400 11400 262	0
0 0 133	264 11500 11500 264	264 11500 11500 264	0
0 0 134	266 11600 11600 266	266 11600 11600 266	0
0 0 135	268 11700 11700 268	268 11700 11700 268	0
0 0 136	270 11800 11800 270	270 11800 11800 270	0
0 0 137	272 11900 11900 272	272 11900 11900 272	0
0 0 138	274 12000 12000 274	274 12000 12000 274	0
0 0 139	276 12100 12100 276	276 12100 12100 276	0
0 0 140	278 12200 12200 278	278 12200 12200 278	0
0 0 141	280 12300 12300 280	280 12300 12300 280	0
0 0 142	282 12400 12400 282	282 12400 12400 282	0
0 0 143	284 12500 12500 284	284 12500 12500 284	0
0 0 144	286 12600 12600 286	286 12600 12600 286	0
0 0 145	288 12700 12700 288	288 12700 12700 288	0
0 0 146	290 12800 12800 290	290 12800 12800 290	0
0 0 147	292 12900 12900 292	292 12900 12900 292	0
0 0 148	294 13000 13000 294	294 13000 13000 294	0
0 0 149	296 13100 13100 296	296 13100 13100 296	0
0 0 150	298 13200 13200 298	298 13200 13200 298	0
0 0 151	300 13300 13300 300	300 13300 13300 300	0
0 0 152	302 13400 13400 302	302 13400 13400 302	0
0 0 153	304 13500 13500 304	304 13500 13500 304	0
0 0 154	306 13600 13600 306	306 13600 13600 306	0
0 0 155	308 13700 13700 308	308 13700 13700 308	0
0 0 156	310 13800 13800 310	310 13800 13800 310	0
0 0 157	312 13900 13900 312	312 13900 13900 312	0
0 0 158	314 14000 14000 314	314 14000 14000 314	0
0 0 159	316 14100 14100 316	316 14100 14100 316	0
0 0 160	318 14200 14200 318	318 14200 14200 318	0
0 0 161	320 14300 14300 320	320 14300 14300 320	0
0 0 162	322 14400 14400 322	322 14400 14400 322	0
0 0 163	324 14500 14500 324		

determined for 20  $\alpha_1$  reflections in the region  $142^\circ \leq 2\theta \leq 161^\circ$  with the  $2\theta$ -scan technique. The averages of these were fitted by least-squares calculations to the Nelson-Riley (1954) function and gave  $a_0 = 24.154 \pm 0.001 \text{ \AA}$  [ $\lambda(\text{Cu } K\alpha_1) = 1.54051 \text{ \AA}$ ].

The extreme reactivity of Na<sub>6</sub>Tl precluded experimental determination of the density. The unit-cell content was estimated with the use of average atomic volumes assuming a contraction of 10%. Assumption of 239  $\text{\AA}^3$  per formula unit of Na<sub>6</sub>Tl leads to a unit-cell content of approximately 412 atoms.

#### Intensity data

Intensities were measured on a locally assembled, Syntex-automated, E&A full-circle diffractometer, equipped with a scintillation detector modified according to Samson (1966), an Ortec counter circuitry with pulse-height analyser, a graphite monochromator designed and built in this laboratory's instrument shop, and a molybdenum-target X-ray tube. The diffractometer was aligned according to Samson (1967a). The X-ray collimator and the counter apertures were designed locally so as to provide maximum signal-to-background ratios.

The  $2\theta$ : $\theta$ -scan method was used throughout at scanning speeds proportional to the peak intensities between 200 and 1500 counts per sec. The scanning speed used at, or below, 200 cps (peak counts) was  $0.5^\circ \text{ min}^{-1}$  and at, or above, 1500 cps  $2^\circ \text{ min}^{-1}$ . The total background counting time for each reflection was equal to the scanning time. This variable-scan-speed program was written by Dr Richard Stanford for the on-line Varian 620i computer controlling the diffractometer. This is the same principle now applied in a refined fashion to the well-known Syntex diffractometer.

The counter-aperture-to-crystal distance was set so as to give an optimal signal-to-background ratio, which was established by plotting the distance against

the ratio. At the optimal setting of 3.25 inches the ratio was improved by a factor of four as compared to that obtained at the distance of about 1.25 inches (or less) used on certain unmodified commercial instruments. The locally designed X-ray tube holder used here provides for both horizontal and vertical tube alignment. In the present experiment, the X-ray tube was set vertically, and the plane-normal of the graphite monochromator was perpendicular to the diffraction vector.

The crystal used was a sphere of 0.117 mm radius. Spherical absorption-correction factors for  $\mu R = 1.84$  were applied to the measured intensities, and the effect of the monochromator on the polarization was taken into account with the use of the formula given by Arndt & Willis (1966), inserting  $\theta_M = 6.08^\circ$  and  $\epsilon = 90^\circ$ . 1602 symmetrically independent reflections were measured, of which as many as 989 had an integrated intensity which was less than, or equal to, one standard deviation. The large number of very weak reflections is due to the high temperature factor associated with the low-melting point and the softness of the compound. Some very low-order reflections were feared to be affected by air scattering because of their proximity to the primary beam and were given zero weight. All other  $F_o$  values marked in Table 1 by two dashes were given zero weight because the integrated intensity was either zero or negative.

#### Derivation and isotropic refinement of the structure

The structure was derived by the method outlined by Samson (1964) which employs packing maps of the 'most useful planes', the nature of which depends on the space group. Since the crystal of Na<sub>6</sub>Tl was assumed (and soon found) to have the space-group symmetry  $F\bar{4}3m$ , many of the arguments used to determine its atomic arrangement are the same as those in the case of Cu<sub>4</sub>Cd<sub>3</sub> (Samson, 1967b), and the reader is

Table 2. *The refined positional parameters ( $\times 10^5$ ), population factors, and isotropic temperature factors for Na<sub>6</sub>Tl*

The standard deviations are given in parentheses.

Atom no.	Kind	Point set	<i>x</i>	<i>z</i>	<i>B</i>
1	Na	48( <i>h</i> ) <i>xxz</i> , etc.	14228 (14)	03346 (21)	5.7 (2) $\text{\AA}^2$
2	Na	48( <i>h</i> ) <i>xxz</i> , etc.	08984 (17)	26791 (20)	6.3 (2)
3	Na	48( <i>h</i> ) <i>xxz</i> , etc.	15395 (16)	52856 (19)	6.4 (3)
4	Na	48( <i>h</i> ) <i>xxz</i> , etc.	05604 (15)	76942 (21)	5.9 (2)
5	Na	48( <i>h</i> ) <i>xxz</i> , etc.	19809 (14)	90985 (20)	5.9 (2)
6	Na	24( <i>f</i> ) <i>x00</i> , etc.	10858 (28)		5.2 (3)
7	Tl	24( <i>f</i> ) <i>x00</i> , etc.	33516 (4)		4.39 (4)
8	Na*	24( <i>g</i> ) $x\frac{1}{4}\frac{1}{4}$ , etc.	06815 (43)		7.3 (6)
9	Na	16( <i>e</i> ) <i>xxx</i> , etc.	30267 (23)		4.1 (4)
10	Na	16( <i>e</i> ) <i>xxx</i> , etc.	41542 (26)		3.4 (4)
11	Na	16( <i>e</i> ) <i>xxx</i> , etc.	55549 (24)		3.6 (3)
12	Na	16( <i>e</i> ) <i>xxx</i> , etc.	67234 (24)		3.6 (3)
13	Tl	16( <i>e</i> ) <i>xxx</i> , etc.	16755 (3)		3.59 (4)
14	Tl	16( <i>e</i> ) <i>xxx</i> , etc.	90136 (3)		3.30 (3)
15	Void†	4( <i>d</i> ) $\frac{1}{4}\frac{1}{4}\frac{1}{4}$ , etc.			

\* Occupancy  $94 \pm 2\%$ .

† Occupancy  $19 \pm 5\%$  if *B* is assumed to be 5.0.

referred to that paper for more details. Space group  $F43m$  has a single most useful plane: the (110) plane.

An enlightening feature discovered in the course of this work was the ease with which the Patterson vectors could be interpreted when they were plotted on a transparent template and then superimposed on the packing map. In such a manner, the only reasonable interpretation of the  $uvw$  Patterson section of  $\text{Na}_6\text{Tl}$  was that the thallium atoms are located at the points marked 7, 13, and 14 on the (110) packing map shown in Fig. 1. It is seen that these points are arranged approximately around a fivefold axis of symmetry. Since the thallium atoms are smaller in radius (by about 10%) than the sodium atoms, it suggested itself that each of these represents the center of an icosahedron. Hence, the basic building block of the structure of  $\text{Na}_6\text{Tl}$  should incorporate a set of five icosahedra arranged about a fivefold axis of symmetry in a manner similar to that observed for one of the four atom complexes occurring in  $\text{Cu}_4\text{Cd}_3$ . Accordingly, the five polygonal sections through the icosahedra marked 7, 13, and 14 in Fig. 1 were placed so as to match the pattern created by the polygonal sections marked 20, 27, and 28 in Fig. 1 of Samson's (1967*b*) paper on  $\text{Cu}_4\text{Cd}_3$ . The perfect geometrical fit of the fivefold sets on the packing map aroused immediate confidence as

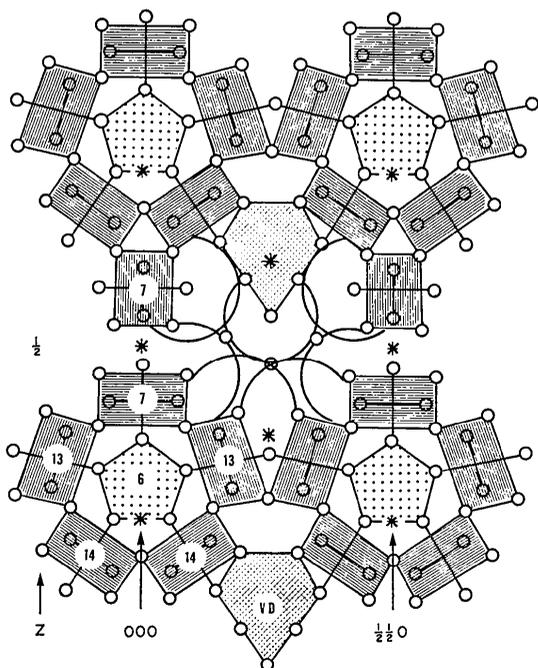


Fig. 1. Packing map of the structure of  $\text{Na}_6\text{Tl}$ , representing the 'most useful plane' [(110) plane] for space group  $F43m$ . The asterisks indicate the location of the fourfold inversion centers. The polygonal sections marked 7, 13, and 14 represent icosahedra arranged approximately about fivefold axes of symmetry. Comparison of this figure with Fig. 1 in the paper describing  $\text{Cu}_4\text{Cd}_3$  (Samson 1967*b*) shows that the  $\text{Na}_6\text{Tl}$  structure type represents part of the structural motif of  $\text{Cu}_4\text{Cd}_3$ .

to the likelihood of the correctness of this model. One more atom (Na 12) had to be added out from the center of each hexagon of the Laves–Friauf polyhedron and possibly one additional atom at the center of each such polyhedron (point set 4(*d*), marked *VD* for void in Fig. 1), which appeared to be somewhat crowded.

This distribution of atoms, which corresponds to the one given in Table 2, accounts for one formula unit of  $\text{Na}_{356}\text{Tl}_{56}$  or  $\text{Na}_{352}\text{Tl}_{56}$  (if 4(*d*) is empty) per unit cube. This agrees well with the content of about 412 atoms expected on the basis of average atomic volumes discussed in the preceding section.

With the atoms placed according to the 15 different point sets given in Table 2, the structure model was tested and refined by least-squares calculations including the 19 positional parameters, the 15 isotropic temperature factors, and the scale factors in one  $35 \times 35$  matrix.

The calculations were done with the use of the *CRYM* system on an IBM 360/75 computer. The quantity minimized was  $\sum_w(F_o^2 - S^2F_c^2)^2$ , where  $1/S$  is the scale factor of  $F_o$  and the weighting scheme is that described by Peterson & Levy (1957), modified so as to apply to  $F^2$  rather than to  $F$ . All the  $R$  values quoted below are defined as  $\sum||F_o| - |F_c||/\sum|F_o|$ . The atomic scattering factors for Na and Tl were taken from Cromer & Waber (1965) and the anomalous dispersion correction factors from Cromer (1965). These were applied according to the scheme given by Dauben & Templeton (1955).

The first refinement was done using intensity data obtained from an irregular crystal at the Ames Laboratory. The refinement converged rapidly, resulting in an  $R$  of 0.16. The center of the Laves–Friauf polyhedron was found to be unoccupied as evidenced by a very high temperature factor for the assumed atom Na (15) (see Table 2).

As the new data became available, the refinement was first continued using all the 615 reflections having net intensities  $I$  larger than  $\sigma(I)$ . Evidence was obtained, again, that the point set 4(*d*) was occupied to only a small extent, or was perhaps empty, and the agreement index  $R$  was 0.064. Inclusion of all the 1602 intensity data in the refinement did not result in a significant change of the positional parameters, but in a noticeable improvement of the standard deviations, and a slight increase in some of the temperature factors.

In the final refinement cycles employing all 1602 data, two population parameters  $p$  were included, one for Na(8), which had a comparatively high temperature factor, and one for point set 4(*d*) [void (15)]. The temperature factor for (4*d*) was fixed somewhat arbitrarily at  $B=5.0$  and excluded from the refinement. In addition, a secondary-extinction parameter  $g$  was applied according to equation (3) of the paper by Larson (1967), the matrix now being  $37 \times 37$ . The largest final shift was close to one third of its standard deviation.

The final  $R$  value obtained with all data included was 0.137, which is considered reasonable in view of

the exceptionally large number of very weak reflections and the scattering from the capillary. The final goodness-of-fit was 1.12, and the extinction parameter refined to  $g = (3.53 \pm 0.15) \times 10^{-9}$ .

The final positional parameters, the population factors, and the isotropic temperature factors are listed in Table 2 and the structure factors in Table 1. Because of the close coupling with the temperature factors, the population parameters may be of questionable accuracy.

The chemical formula that corresponds to the refined model may be taken as  $\text{Na}_{352}\text{Tl}_{56}$  or  $\text{Na}_{6.29}\text{Tl}$  ( $\rho_{\text{calc}} = 2.302 \text{ g.cm}^{-3}$ ). Because of the possibility of variable composition, especially in possible isostructural compounds (see below), it seems practical to retain the name  $\text{Na}_6\text{Tl}$  for this structure type.

### Cursory anisotropic refinement

The relatively large magnitude of the isotropic temperature factors is most likely due to the low melting point of the compound (78°C; see Introduction), which, in turn is reflected in the softness of the crystals. In order to obtain some idea about the anisotropy of the thermal motions, three additional least-squares calculations were carried out, this time with the use of program *XFLS*.\*

The parameters listed in Table 2 were used as a starting point. Full occupancy was assumed of all positions except for 'void 15', which was taken to be occupied 19% of the time by sodium, with  $B(\text{iso})$  fixed at 5.0 (as before). The result is shown in Table 3.

The positional parameter  $X$  of Na(2) changed from 0.08984 (Table 2) to 0.09126, the parameter  $Z$  of Na(3) changed from 0.52856 to 0.52985, and the parameter  $X$  of Na(12) from 0.67234 to 0.67521. The changes in the other positional parameters were equal to, or less than three standard deviations.

\* *XFLS* is an IBM-360 version of *ORFLS* (Busing, Martin & Levy, 1962).

Table 3. Anisotropic thermal parameters for  $\text{Na}_6\text{Tl}$

The expression used was of the form:  $-[B'_{11}h^2 + B'_{22}k^2 + B'_{33}l^2 + 2B'_{12}hk + 2B'_{13}kl + 2B'_{23}kl]$ . The standard deviations are given in parentheses, and  $B'_{ij} = 4a_0^2 B_{ij}(\text{\AA}^2)$ .

	$B_{11}$	$B_{22}$	$B_{33}$	$B_{12}$	$B_{13}$	$B_{23}$
Na(1)	5.9 (2)	$B_{11}$	4.1 (4)	0.3 (3)	-1.1 (2)	$B_{13}$
Na(2)	6.3 (3)	$B_{11}$	6.7 (5)	0.1 (4)	0.8 (3)	$B_{13}$
Na(3)	5.8 (3)	$B_{11}$	6.7 (5)	1.1 (4)	-0.5 (3)	$B_{13}$
Na(4)	5.4 (2)	$B_{11}$	5.5 (5)	-0.6 (3)	0.7 (2)	$B_{13}$
Na(5)	4.8 (2)	$B_{11}$	8.0 (5)	-0.7 (3)	0.3 (2)	$B_{13}$
Na(6)	6.7 (5)	4.2 (3)	$B_{22}$	0	0	-0.1 (5)
Tl(7)	2.96 (5)	4.92 (5)	$B_{22}$	0	0	-0.19 (7)
Na(8)	7.9 (9)	7.5 (5)	$B_{22}$	0	0	0.5 (7)
Na(9)	4.5 (4)	$B_{11}$	$B_{11}$	-0.8 (3)	$B_{12}$	$B_{12}$
Na(10)	4.8 (4)	$B_{11}$	$B_{11}$	0.4 (3)	$B_{12}$	$B_{12}$
Na(11)	4.4 (3)	$B_{11}$	$B_{11}$	0.7 (4)	$B_{12}$	$B_{12}$
Na(12)	5.0 (3)	$B_{11}$	$B_{11}$	2.2 (4)	$B_{12}$	$B_{12}$
Tl(13)	3.56 (2)	$B_{11}$	$B_{11}$	0.37 (5)	$B_{12}$	$B_{12}$
Tl(14)	3.20 (2)	$B_{11}$	$B_{11}$	0.03 (2)	$B_{12}$	$B_{12}$

The  $R$  value for the 615 reflections having net intensities  $I$  larger than  $\sigma(I)$  decreased from 0.064 to 0.050. The overall  $R$  value, for 1599 reflections, including 312 data representing zero intensity, is 0.186.

Since the program has no provision for refining the extinction parameter  $g$  discussed in the preceding section, the two strongest reflections, 066 & 228, were excluded from the refinement.

It is seen that the atoms Na(5) and Na(12) constituting the Laves-Friauf polyhedron show the most pronounced anisotropy.

### Description of the structure

Each of the 56 thallium atoms is surrounded by an icosahedron of 12 sodium atoms. The structure can be described most conveniently in terms of a complex consisting of 14 such icosahedra as shown in Fig. 2. Five icosahedra are arranged approximately about a fivefold axis of symmetry [Fig. 2(a)], thus enclosing a pentagonal prism. These icosahedra are the ones that have been marked 7, 13, and 14 on the packing map Fig. 1. Six such fivefold rings interpenetrate one another and share icosahedra so as to form the aggregate shown in Fig. 2(c), which consists of 14 icosahedra that enclose six pentagonal prisms of the kind shown at the center of Fig. 2(a). The center of each such prism is located at the vertex of an octahedron of  $T_d$  symmetry. Fig. 2(b) shows two such fivefold rings interpenetrating at right angles. It is now seen that the pentagonal prism at the center of Fig. 2(a) is shared between two icosahedra, one above and the other below the plane of the paper. The two icosahedra have one vertex in common at the center of that pentagonal prism, and each icosahedron center is at an extended pole of that prism. Each additional vertex that is shared between two icosahedra represents the center of a pentagonal prism (which has two atoms at its extended poles), as seen in Fig. 2(c). Accordingly, 36 more pentagonal prisms are created.

The aggregate shown in Fig. 2(c), accordingly, represents 14 icosahedra and 42 pentagonal prisms,

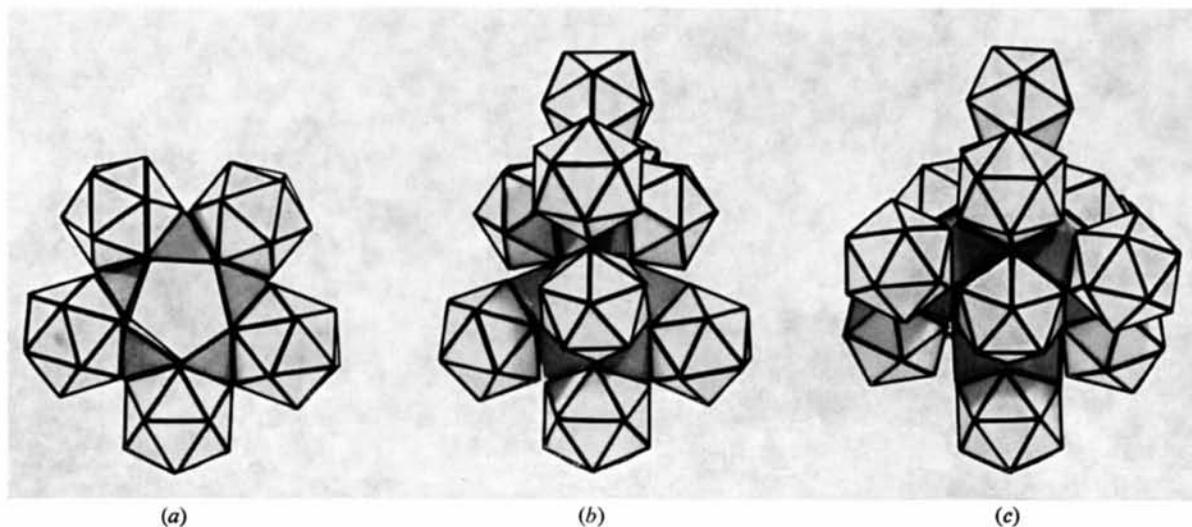


Fig. 2. (a) The basic building block of  $\text{Na}_6\text{Tl}$  consisting of an aggregate of five icosahedra arranged approximately about a fivefold axis of symmetry. (b) Two such fivefold rings interpenetrating at right angles in such a way that the central pentagonal prism in (a) is shared by two icosahedra. (c) Six interpenetrating fivefold rings forming a complex of 14 icosahedra and 42 centered pentagonal prisms.

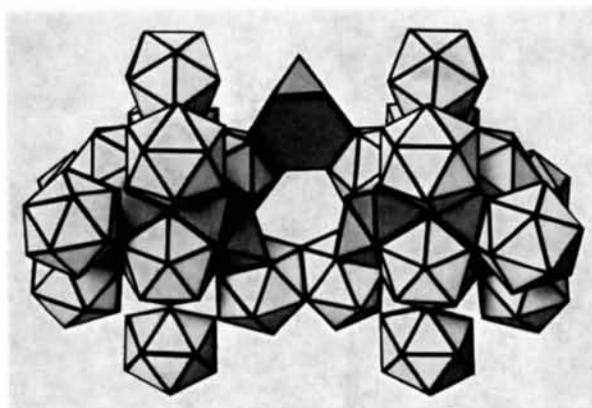


Fig. 3. Two icosahedral complexes of the kind shown in Fig. 2(c) sharing one equilateral triangle each with a Laves-Friauf polyhedron (dark), the center of which is a fourfold inversion center. Hence, each Laves-Friauf polyhedron is shared between four such complexes which are tetrahedrally arranged around the  $\bar{4}$  center. For the sake of perspicuity one icosahedron has been removed from each 98-atom complex.

each one with two atoms at the extended poles. It is almost identical to one of the four aggregates observed in the structure of  $\text{Cu}_4\text{Cd}_3$  (Samson, 1967*b*), which, however, contains 1124 atoms per smallest unit cube.

The unit cell of  $\text{Na}_6\text{Tl}$  contains four aggregates of the kind shown in Fig. 2(c). These are arranged about the points, 0,0,0, etc. [point et 4(a)], and share edges and faces in such a way that the average number of atoms per aggregate is 98. Thus, the four aggregates account for 392 atoms per smallest unit cube. With the addition of 16 more atoms [16 Na(12)], each one out from the center of a hexagon of a Laves–Friauf polyhedron, the entire complement of 408 atoms in the unit cell is accounted for.

Fig. 3 shows two icosahedral (98-atom) complexes, each one sharing a triangle with the Laves–Friauf polyhedron (dark) the center of which coincides with a fourfold inversion center. Thus it is seen that each Laves–Friauf polyhedron is shared between four icosahedral complexes that are tetrahedrally arranged around a  $\bar{4}$  center. As has already been noted, the center of the Laves–Friauf polyhedron ('void' in Table 2) is probably not occupied.

#### Other complex cubic $\text{A}_6\text{B}$ compounds

The crystal-structure studies at this laboratory have revealed, so far, that there exist at least two types of complex cubic  $\text{A}_6\text{B}$  compounds with similar unit-cell contents: the  $\text{Na}_6\text{Tl}$  type containing 408 atoms per structural unit and the  $\text{Mg}_6\text{Pd}$  type (Samson, 1972) containing 396 atoms. Further structural details, including the interatomic distances, which account for similarities as well as the differences between the two types are discussed in the paper following this one (Samson, 1972).

Since the refined structure model does not differ from the original one communicated to the Ames Laboratory, and the shortest Tl–Tl distance observed

is 5.631 Å, the general structural features of  $\text{Na}_6\text{Tl}$  discussed by Greiner, *et al.* (1969) are still valid.

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